

# **Oxidation Behavior of Surface-modified Stainless Steel 316LN in Supercritical-CO<sub>2</sub> Environment**

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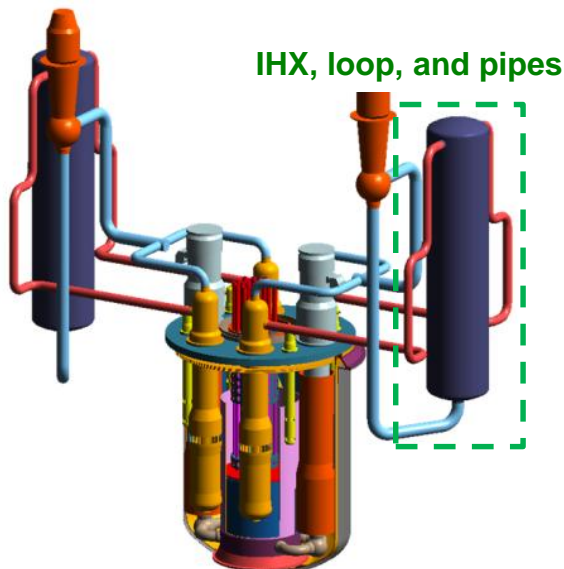
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## ❑ Supercritical-CO<sub>2</sub> (S-CO<sub>2</sub>) Brayton cycle

- Cross-cutting power conversion technology
  - Sodium-cooled Fast Reactor (SFR)
    - ❑ Superheated steam Rankine → S-CO<sub>2</sub> Brayton cycle at 500-550°C (20MPa)

## ❑ Material issues

- Oxidation and carburization in S-CO<sub>2</sub> environment
  - Loss of load bearing area
  - Integrity of heat exchanger and turbomachinery
  - Mechanical property degradation



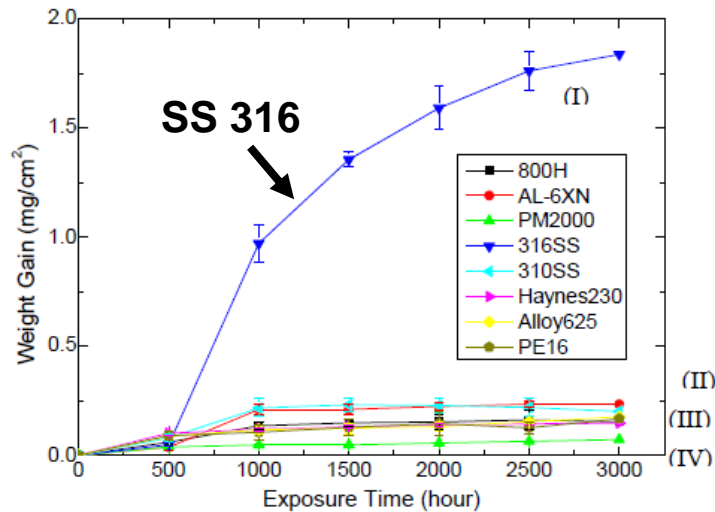
### Prototype Gen IV Sodium-cooled Fast Reactor (PGSFR)

- Pool-type reactor (150MWe)
  - Core in/out temperature → 390/545°C
  - Liquid sodium coolant
  - Superheated steam Rankine cycle
- S-CO<sub>2</sub> cycle considered as replacement

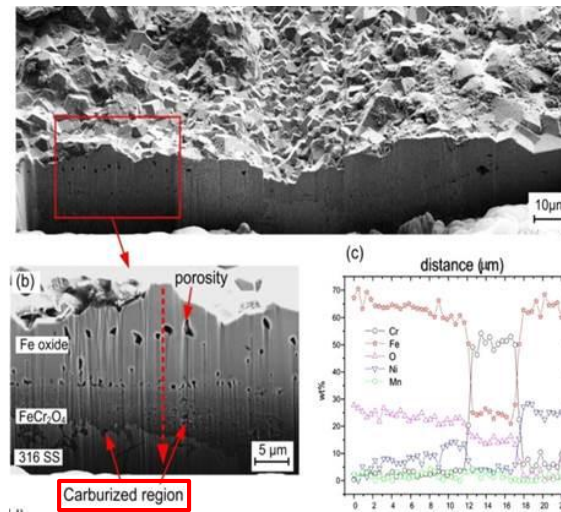
▲ Schematic image and operating conditions of PGSFR

## ❑ Stainless steel 316LN

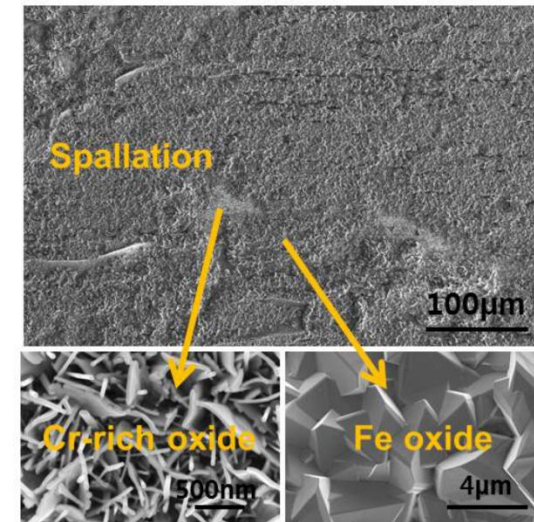
- Candidate material for key components and structures in SFR
- Comparatively poor oxidation behavior in S-CO<sub>2</sub> environment  
 → Other various candidate materials considered
- Cr-rich oxide spallation followed by **thick Fe-oxide** formation  
 → **Necessary to improve oxidation behavior of SS 316LN**



▲ Weight gain of candidate materials after corrosion in S-CO<sub>2</sub> at 650°C (20MPa) up to 3000h [1]



▲ Corrosion behavior of 316SS in S-CO<sub>2</sub> at 650°C (20MPa) for 3000h [2]

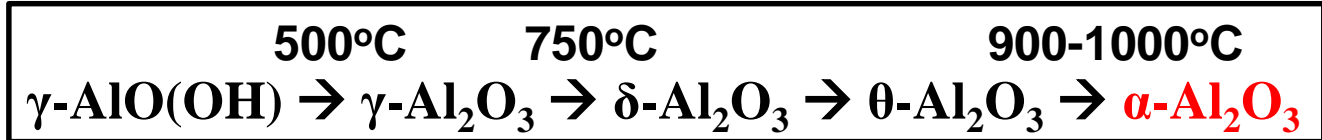


▲ SEM micrograph of oxide surface of SS 316LN after exposure in S-CO<sub>2</sub> at 650°C (20MPa) for 1000h [3]

# Introduction

## Alumina ( $\text{Al}_2\text{O}_3$ )

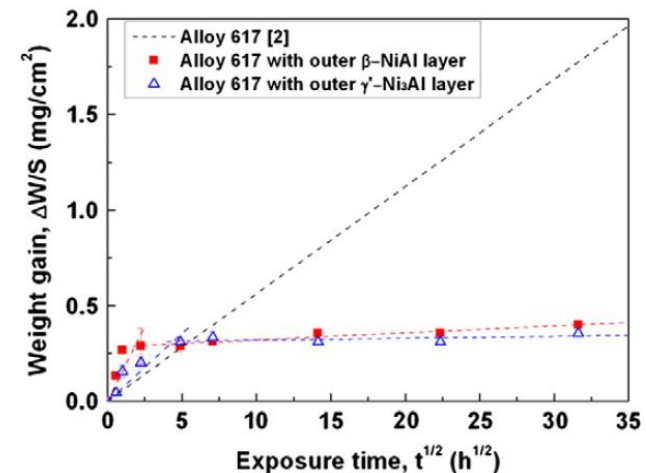
- Various transitional phases depending on temperature



- $\alpha\text{-Al}_2\text{O}_3$ : superior oxidation and carburization resistance
- Al content over 4-5 wt.% for continuous  $\text{Al}_2\text{O}_3$  layer formation

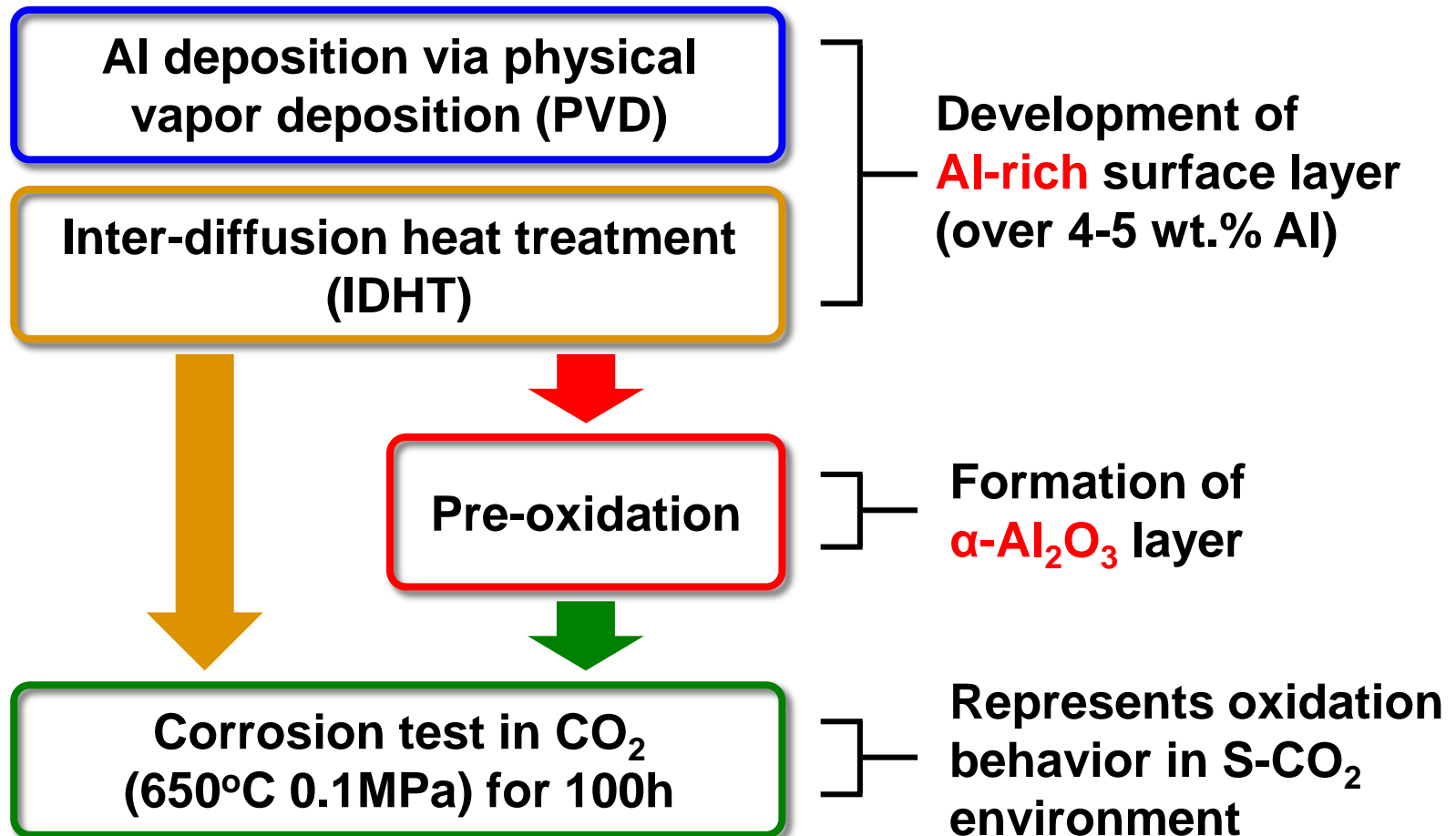
## Surface-modification

- Diffusion coating of SS 316LN with Al
  - Surface enrichment  $\rightarrow$  promote  $\text{Al}_2\text{O}_3$  formation
- Past work on Ni-base superalloy
  - $\beta\text{-NiAl}$  surface layer formation
  - Oxidation in air (900°C 1000h)
    - Transient oxidation behavior  $\rightarrow$   $\text{NiAl}_2\text{O}_4$ ,  $\alpha\text{-Al}_2\text{O}_3$



▲ Weight gain curves of as-received and IDHTed Alloy 617 oxidized in air (900°C) up to 1000h [1]

- Improvement of oxidation resistance of SS 316LN in S-CO<sub>2</sub> environment via surface-modification



## □ Surface-modification

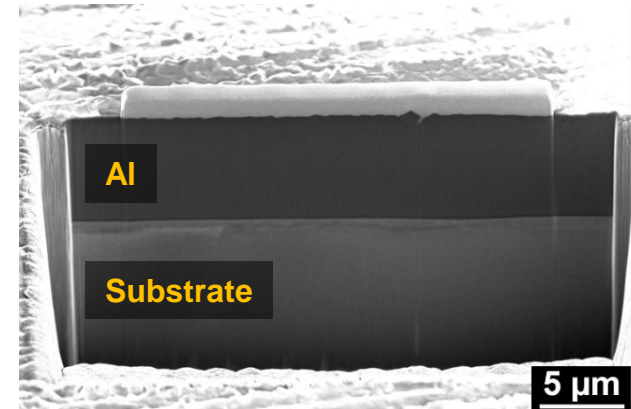
- Al coating via PVD

Target	Base pressure	Working pressure (Ar)	RF power	Duration	Coating Thickness
Al (99.999%)	$\leq 2.0 \times 10^{-6}$ torr	$0.7 \times 10^{-4}$ torr	200 W	4 h	~ 8 $\mu\text{m}$

- IDHT: 600°C 0.5h + 1000°C 0.5h (vacuum,  $\leq 2.0 \times 10^{-6}$  torr)
- Pre-oxidation: 900°C 1h (He, 99.999% purity)

## □ CO<sub>2</sub> corrosion test

- Once-through type quartz, three-zone heater
- CO<sub>2</sub>: 99.999% purity with flow rate of 150 sccm
- Exposure to CO<sub>2</sub> at 650°C 0.1MPa for 100h



▲ Cross-sectional SEM micrograph of Al-coated SS 316LN

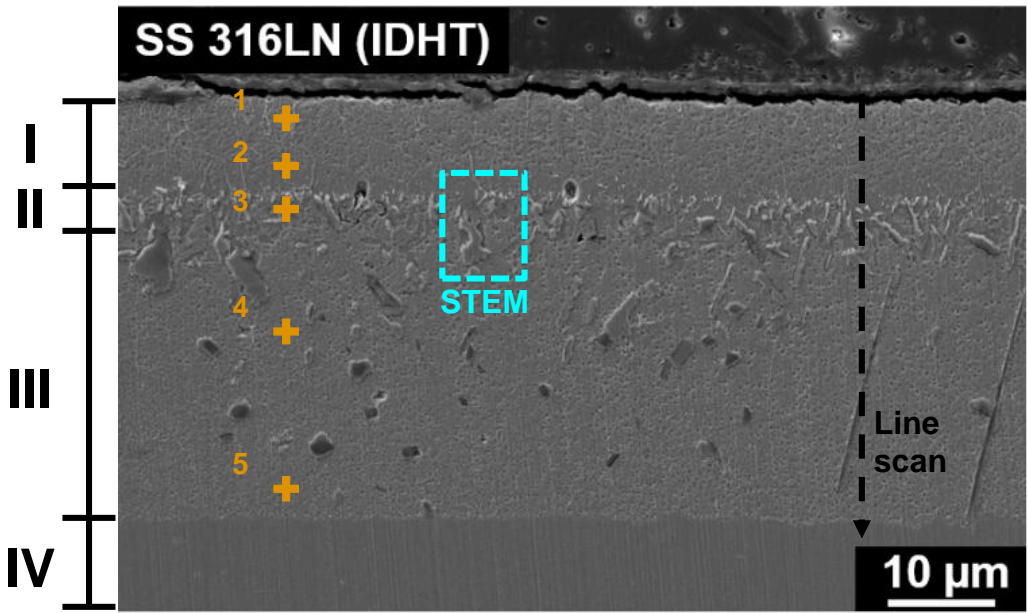
▼ Chemical composition of SS 316LN used in this study (in wt.%)

Material	Fe	Cr	Ni	C	Mo	Mn	Si	N
SS 316LN	Bal.	18.9	13.9	0.03	2.8	1.9	0.6	0.16

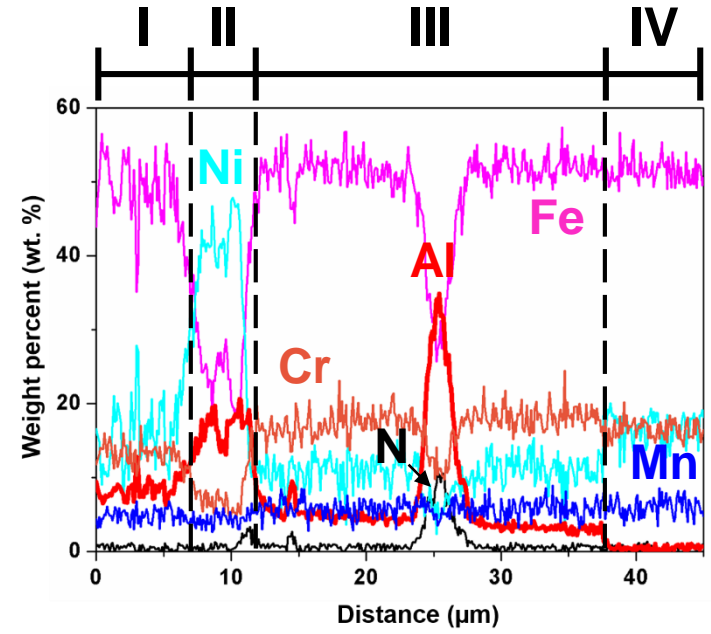
# Results and discussion

## Microstructure of IDHTed SS 316LN

- Al-rich surface layer formed as result of inter-diffusion
  - Al content generally above 5 wt.% in aluminized region
- Distinct regions
  - I: Al-rich surface area (Al > 10 wt.%)
  - II: Ni-Al rich intermetallic
  - III: Al-rich area (Al > 4 wt.%) with AlN
  - IV: Substrate



▲ Cross-sectional SEM micrograph of IDHTed SS 316LN



▲ EDS line scan analysis of IDHTed SS 316LN

wt. %	Fe	Cr	Ni	Al
Point 1	41.6	20.2	11.7	12.0
Point 2	34.9	17.8	19.0	15.8
Point 3	17.9	8.7	38.7	24.3
Point 4	44.3	28.7	6.3	5.4
Point 5	40.9	29.4	10.1	4.4

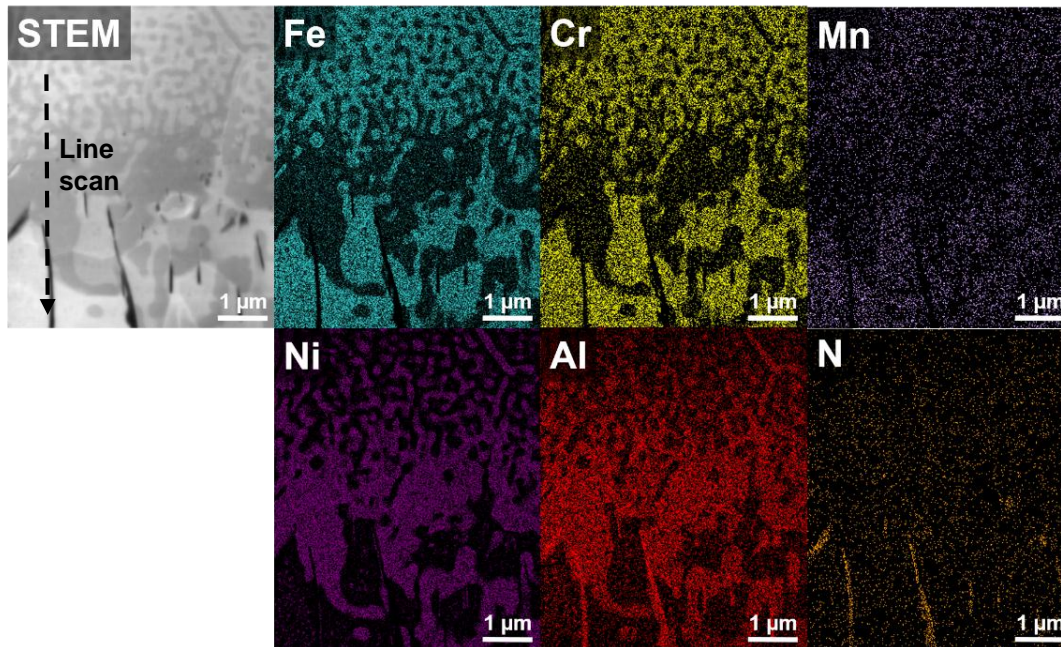
▲ Point EDS analysis of IDHTed SS 316LN



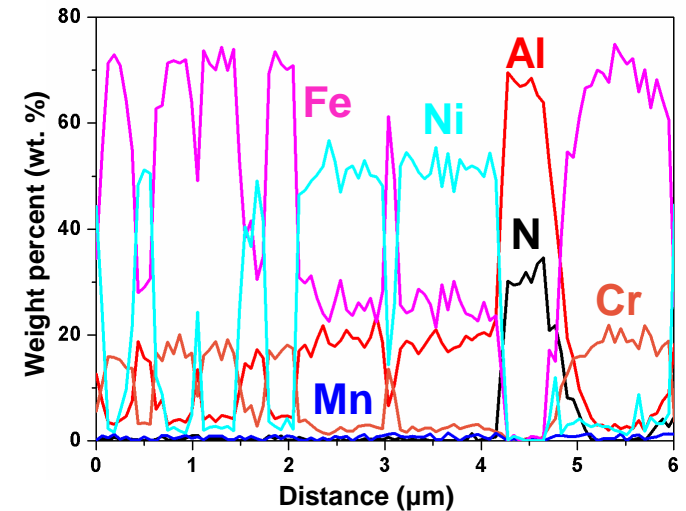
# Results and discussion

## □ Microstructure of IDHTed SS 316LN

- Al preferentially forming intermetallic with Ni → Nickel aluminide ( $\beta$ -NiAl)
- Comparative enrichment of Fe and Cr in Ni-depleted areas



▲ STEM and elemental mapping images of IDHTed SS 316LN

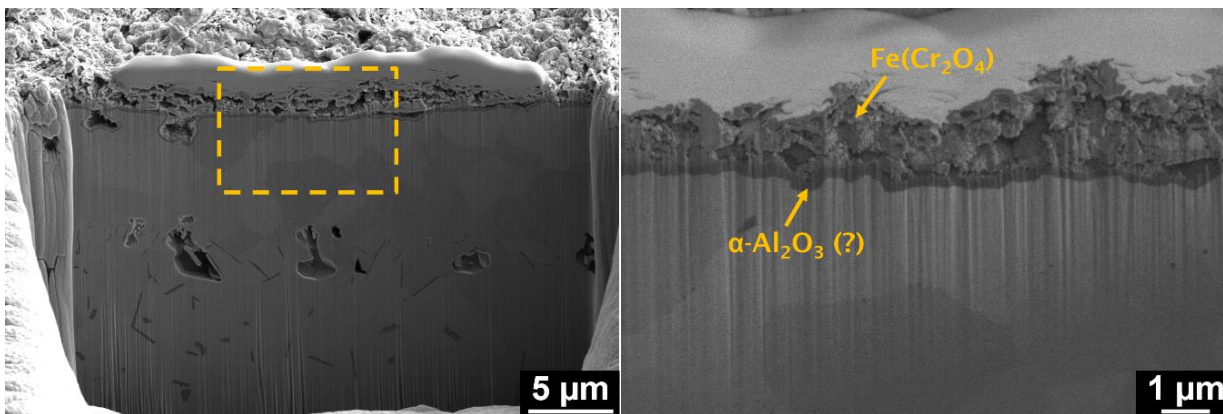


▲ EDS line scan analysis of IDHTed SS 316LN

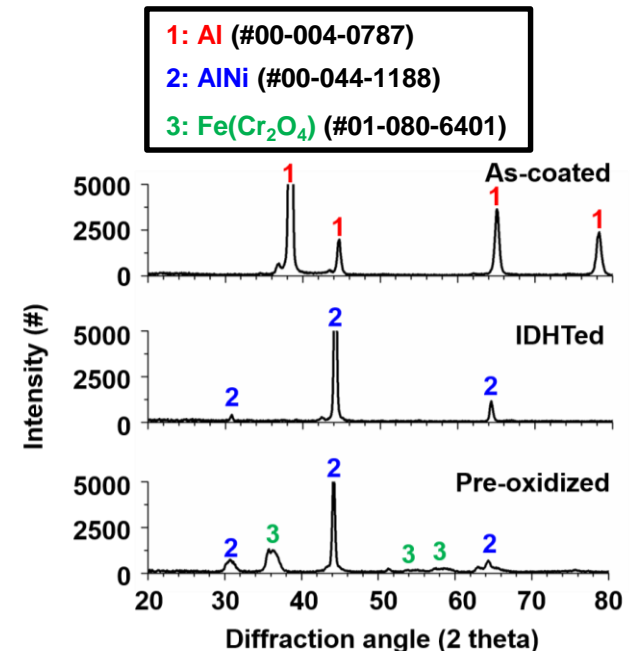
# Results and discussion

## □ Microstructure of pre-oxidized SS 316LN

- Pre-oxidation performed to form  $\alpha\text{-Al}_2\text{O}_3$ 
  - Operational temperature ( $\sim 550^\circ\text{C}$ ) not high enough for formation of  $\alpha\text{-Al}_2\text{O}_3$
- XRD analysis: mainly  $\text{Fe}(\text{Cr}_2\text{O}_4)$   $\rightarrow$  Fe, Cr enriched areas at surface
  - $\alpha\text{-Al}_2\text{O}_3$  peaks undetected  $\rightarrow$  comparatively too thin to be detected
- Cross-sectional BSE micrograph reveal oxide inner layer of different composition than outer layer
  - $\rightarrow$  possibly  $\alpha\text{-Al}_2\text{O}_3$  layer



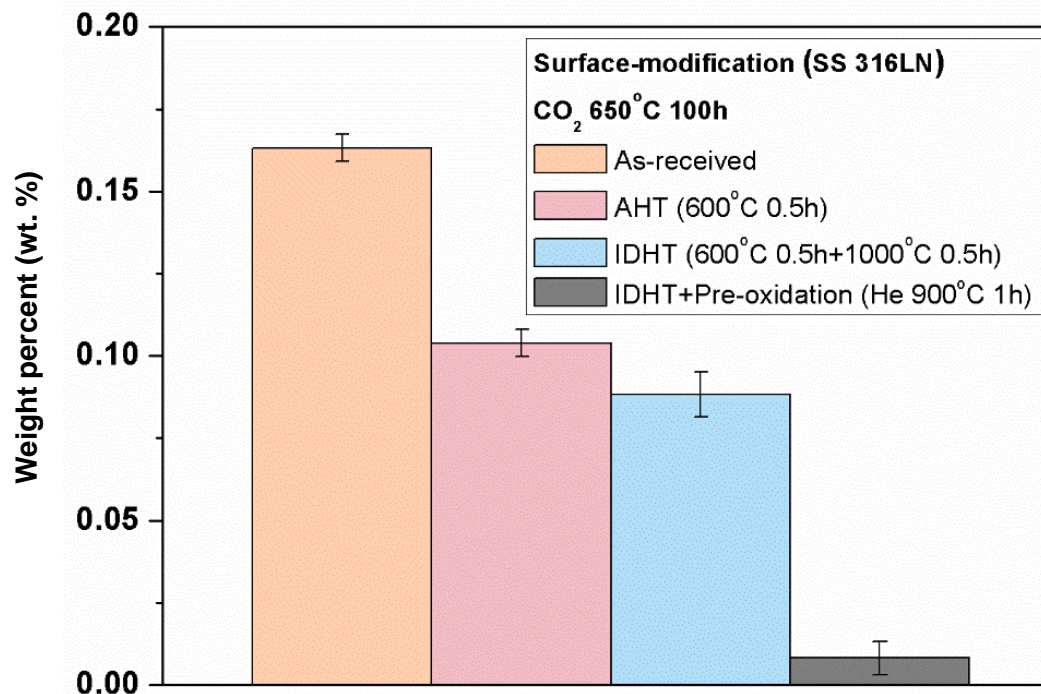
▲ Cross-sectional SE/BSE micrographs of pre-oxidized SS 316LN



▲ XRD analysis of surface-modified SS 316LN

## □ Oxidation behavior in CO<sub>2</sub> environment

- Weight change
  - As-received > IDHTed > Pre-oxidized
  - Necessary to evaluate **extent of pre-oxidation** → weight gain & oxide thickness

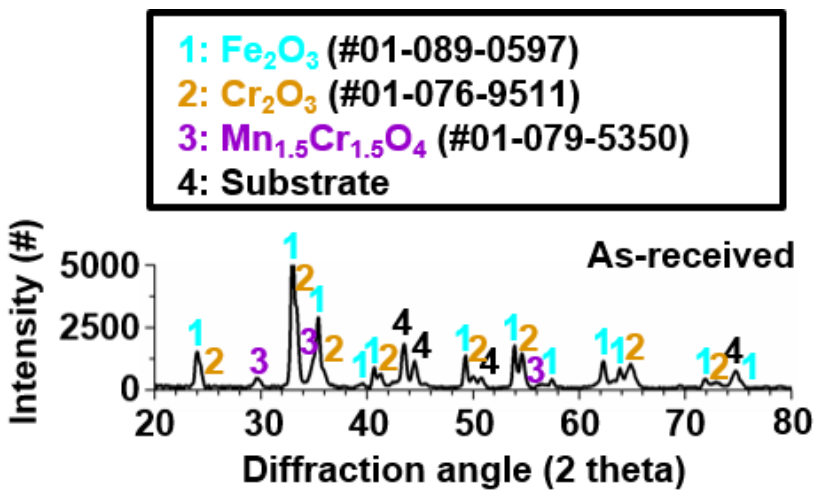


▲ Weight change after exposure to CO<sub>2</sub> (650°C 0.1MPa 100h)

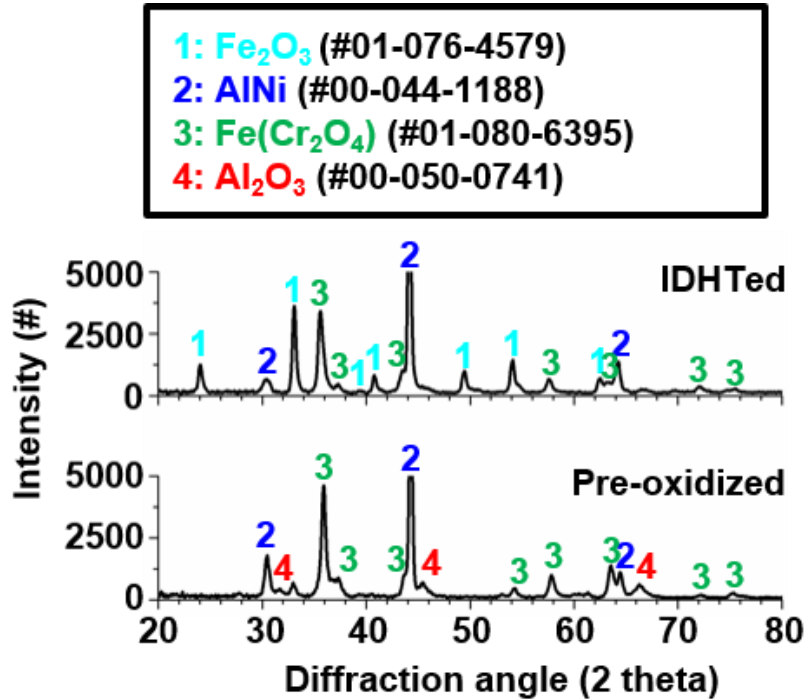
# Results and discussion

## ❑ Oxidation behavior in CO<sub>2</sub> environment

- XRD analysis
  - As-received, IDHTed → Fe<sub>2</sub>O<sub>3</sub> formation
    - Weight gain difference → detailed oxide analysis necessary
  - Pre-oxidized → No Fe<sub>2</sub>O<sub>3</sub>, Al<sub>2</sub>O<sub>3</sub> formation → low weight gain



▲ XRD analysis of as-received SS 316LN after exposure to CO<sub>2</sub> (650°C 0.1MPa 100h)



▲ XRD analysis of surface-modified SS 316LN after exposure to CO<sub>2</sub> (650°C 0.1MPa 100h)

## □ Summary

- SS 316LN was coated with Al and heat-treated (IDHT)
  - Development of Al-rich surface layer (above 4-5 wt.% Al)
  - Nickel aluminide (NiAl), AlN formation
- IDHTed SS 316LN was pre-oxidized
  - Formation of  $\alpha\text{-Al}_2\text{O}_3$  prior to  $\text{CO}_2$  exposure
- After  $\text{CO}_2$  exposure at  $650^\circ\text{C}$  for 100h
  - Weight gain: As-received > IDHTed > Pre-oxidized
    - May be attributed to pre-formation of  $\text{Al}_2\text{O}_3$
  - $\text{CO}_2$  of  $650^\circ\text{C}$  not high enough temperature for formation of protective  $\alpha\text{-Al}_2\text{O}_3$ 
    - ➔ **Pre-oxidation procedure necessary for significantly improved oxidation behavior**

## ❑ Concerns

- Diffusional loss of Al in longer durations
  - Insufficient Al content to maintain stable Al<sub>2</sub>O<sub>3</sub> layer
  - Formation of brittle aluminides in affected bulk
- Carburization resistance

## ❑ Further research

- Detailed analysis of CO<sub>2</sub> (650°C 100h) exposed specimens
- Longer exposure times
- Evaluation of co-deposited Ni, Al coating performance
  - Diffusion barrier for Al at surface
- Corrosion tests in S-CO<sub>2</sub> (20MPa) environment
  - Comparison with CO<sub>2</sub> (0.1MPa) results
- Assessment of carburization resistance
- Mechanical property evaluation

**Energy for Earth !!**



**Thank you!**