# Fabrication of Mo Microplate dispersed UO<sub>2</sub> Nuclear Fuel Pellets

Dong Seok Kim<sup>\*</sup>, Dong-Joo Kim, Jang Soo Oh, Sang-Chae Jeon, Keon Sik Kim, Jong Hun Kim, and Jae Ho Yang Nuclear Fuel Safety Research Division, Korea Atomic Energy Research Institute \*Corresponding author: dskim86@kaeri.re.kr

#### 1. Introduction

Through the decades of developments of nuclear fuel pellets, many of efforts have been focused on increasing the economic efficiency of the LWR power generation such as, increasing the fuel discharged burnup, extending the fuel cycle, and up-rating the maximum power. However, in the wake of Fukushima accident, it becomes more important recently, and well-known that the current LWR fuel should be tolerable to severe accidents to mitigate their consequence with maintaining the performances. Thus, various concepts of new fuels are being suggested and developed under the name of accident tolerant fuels (ATF).

One of the current issues for nuclear UO<sub>2</sub> fuel pellet is about its low thermal conductivity. The low thermal conductivity leads to increase thermal gradient in the fuel pellet and centerline temperature when in operation. Enhancing the thermal conductivity of UO<sub>2</sub> fuel pellet is greatly attractive in the aspect of fuel performance [1-3]and also for its safety margin. The fuel pellets having high thermal conductivity can lower fuel temperature and reduce the mobility of the fission gases [4-6]. In addition, a reduced temperature gradient within the pellet probably enhances the dimensional stability, with lower thermal stress of the fuel pellet, thus the pellet cladding mechanical interaction (PCMI) and even in fuel fragmentation, relocation and dispersal (FFRD) can be mitigated. A thermal margin gained from the high thermal conductivity of pellet would be utilized in a safe operation of LWR or even power-uprate operation also. There have been efforts on enhancing the thermal conductivity of the fuel pellet. One of the methods is introducing high thermal conductive materials into fuel pellets. Yang et al. [7] have shown experimentally that the thermal conductivity of a UO<sub>2</sub> pellet can be increased substantially by providing a UO<sub>2</sub> pellet with connected tungsten channel. KAERI has also developed micro-cell UO<sub>2</sub> fuel pellets consist of granules enveloped by thin metallic cell walls. [8-10] The metallic cell walls in pellets are continuously connected to each other, enhancing thermal conductivity.

In this study, to enhance the thermal conductivity in radial direction, molybdenum metal microplate was dispersed in a  $UO_2$  fuel pellet. Micrometer-sized thin Mo plates were aligned horizontally in a  $UO_2$  pellet to have enhanced thermal conductivity with heat transfer paths in radial direction. Moreover, the compatibility in the fuel fabrication process can be enhanced, due to the simple pellet fabrication method. The thermal properties

of the pellets were characterized with the microstructures of the fuel composite.

### 2. Experimental and Result

A Mo microplate  $UO_2$  pellet was fabricated by composing  $UO_2$  and Mo microplate powders.

Mo microplates were prepared by milling spherical Mo powder particles. Mo metal powder (SIGMA-ALDRICH, 99.9%) was milled in a planetary milling machine and three kinds of Mo platelets were prepared varying particle sizes. Fig. 1 shows Mo platelets prepared in this study. After the milling process, spherical Mo particles were transformed to micro-sized thin plates, having three different size distributions with the prepared powders.

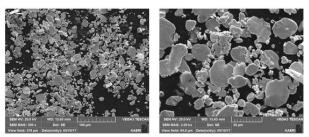


Fig. 1. SEM images of Mo microplates with different magnifications. (500x and 2000x)

5 vol.% of Mo microplates were simply mixed in a tubular mixer with UO<sub>2</sub> powder. The powder mixture was compacted using a uniaxial press at about 300 MPa, and the pelletized green body was sintered at 1730  $^{\circ}$ C for 4h in a flowing H<sub>2</sub> atmosphere.

The sintered density of a Mo microplate pellet was determined using an immersion method, and a microstructure of the sintered pellet was observed using optical microscopy and SEM.

Fig. 2 shows the microstructure of a Mo microplate  $UO_2$  pellet. The bright phase of Mo microplates were dispersed homogeneously and aligned in horizontal direction in  $UO_2$  pellet, which is forming effective thermal conductive paths for radial heat transfer.

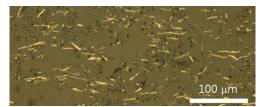


Fig. 2. Microstructure of a Mo microplate UO<sub>2</sub> pellet.

Thermal conductivity of the pellet was characterized by LFA method. The pellet was sliced in axial direction to measure the effective radial thermal conductivity. The radial thermal conductivity was much enhanced compared with bare UO2, and even also higher than the conductivity of the UO<sub>2</sub> pellet with same amount of spherical Mo particles included. (Fig. 3) This enhancement of the thermal conductivity of the Mo microplate UO<sub>2</sub> pellet was mainly affected by the shape and arrangement of the metallic plates in the pellet. The effect on the thermal conductivity with the Mo plates was investigated.

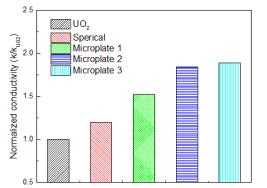


Fig. 3. Comparison of thermal conductivities of  $UO_2$  fuel pellets at 1000°C.

### 3. Summary

In this study, Mo microplate  $UO_2$  nuclear fuel pellet was fabricated for enhancing the thermal conductivity of the pellet. Mo metal microplates were aligned working as heat conducting paths in the pellet. Therefore, the thermal conductivity of the  $UO_2$  pellet in radial direction could be enhanced, which can lead to reduce thermal gradient of the pellet when in operation in a reactor. Considering the outstanding fuel pellet characteristics, this Mo microplate  $UO_2$  pellet will be one of the promising fuel concepts of ATF pellets in near future.

## ACKNOWLEDGEMENT

This work was supported by the National Research Foundation of Korea (NRF) grant funded by the Korean government (MSIT: Ministry of Science and ICT) (No. 2017M2A8A5015056).

## REFERENCES

[1] B.H. Lee, Y.H. Koo, J.S. Cheon, J.Y. Oh, H.K. Joo, and D.S. Sohn, J. Korean Nucl. Soc. 34 (5) (2002) 482–493

[2] B.H. Lee, Y.H. Koo, J.Y. Oh, J.S. Cheon, Y.W. Tahk, and D.S. Sohn, Nucl. Eng. Technol. 43 (6) (2011) 499–508.

[3] A.F. Williams, B.W. Leitch, and N. Wang, Nucl. Eng. Technol. 45 (7) (2013) 839–846.

[4] C.R.A. Catlow, Proc. R. Soc. Lond. Ser. A 364 (1978) 473–499.

[5] K. Forsberg and A.R. Massih, J. Nucl. Mater. 135 (1985) 140–148.

[6] Y.H. Koo, J.Y. Oh, B.H. Lee, Y.W. Tahk, and K.W. Song, J. Nucl. Mater. 405 (2010) 33–43.

[7] J.H. Yang, K.W. Song, K.S. Kim, and Y.H. Jung, J. Nucl. Mater., 353 (2006) 202-208

[8] J.H. Yang, K.S. Kim, D.J. Kim, J.H. Kim, J.S. Oh, Y.W. Rhee, Y.H. Koo, TopFuel 2013, American Nuclear Society, Charlotte, September 15–19, 2013

[9] D.-J. Kim, Y. W. Rhee, J. H. Kim, Y. W. Rhee, D.-J. Kim, K. S. Kim, J. S. Oh, J. H. Yang, Y.-H. Koo and K.-W. Song, J. Nucl. Mater., 462, (2015) 289.

[10] D.-J. Kim, K. S. Kim, D. S. Kim, J. S. Oh, J. H. Kim, J. H. Yang, Y.-H. Koo, Nucl. Eng. Tech., 50 (2018) 253.