

Emulating Neutron Irradiation Effect on Stainless steel by Non-Irradiation Method

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Korean Nuclear Society 2024

Jeju ICC

2024.05.09

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ACKNOWLEDGEMENT

This work was financially supported by National Research Foundation (NRF) of Korea Grant funded by the Ministry of Science and ICT (MSIT) (Grant Number: RS-2022-00144514)

Structure of Nuclear Power Plant

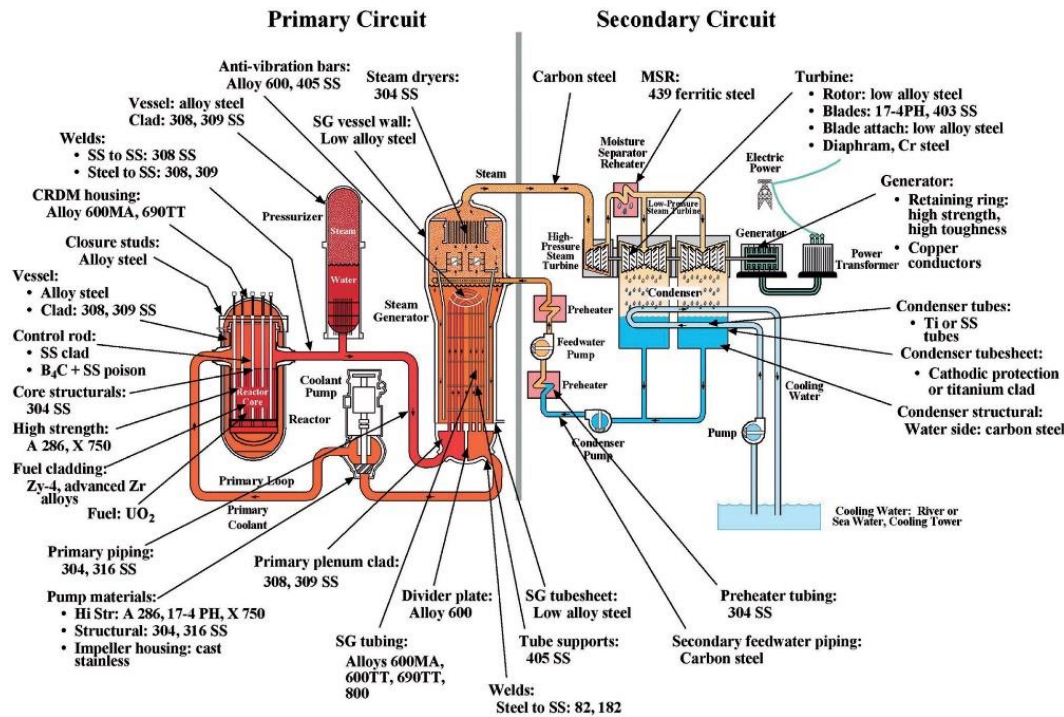


Fig. PWR components and materials [1]

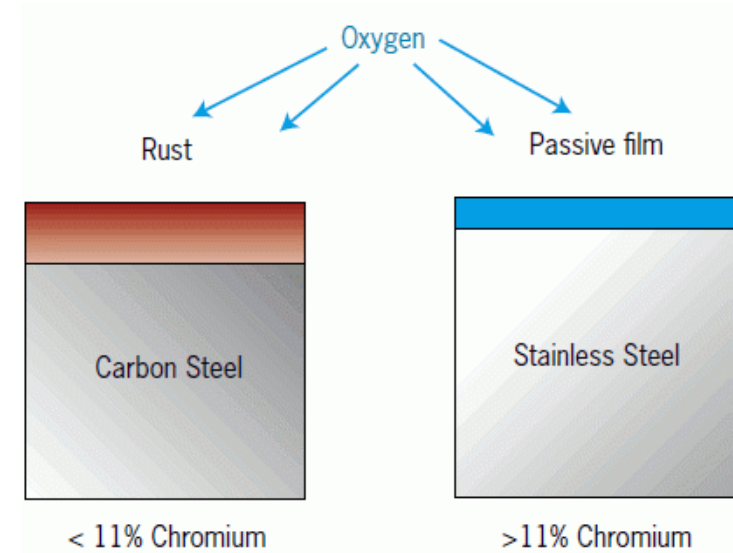


Fig. Difference between ordinary steel and stainless steel

- Austenitic stainless steel, such as 304 and 316, demonstrates superior mechanical properties and corrosion resistance in nuclear reactor environments.
- Compared to conventional steel, stainless steel contains a higher chromium content, which enhances corrosion resistance through the formation of a passive layer.

Effects of Neutron Irradiation on Metals

- Radiation-induced segregation (RIS) modifies the micro composition in regions with high diffusion coefficients, such as grain boundaries.
- Defects arise within the material, resulting in its hardening due to these imperfections.
- This process diverges from conventional thermodynamic material alterations and tendencies, warranting additional research.

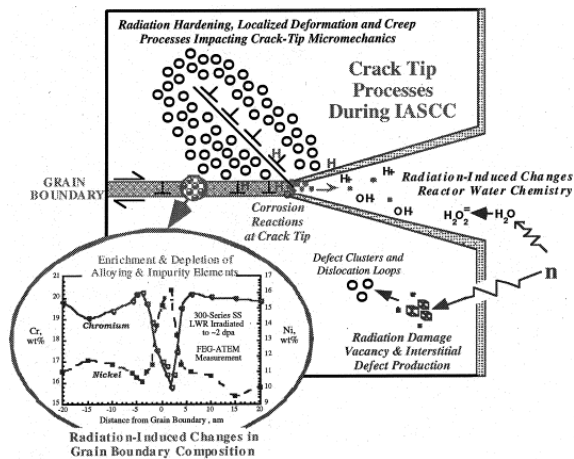


Fig. Radiation induced segregation [1]

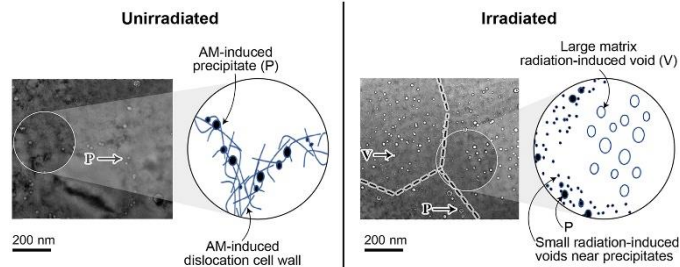


Fig. Defect [2]

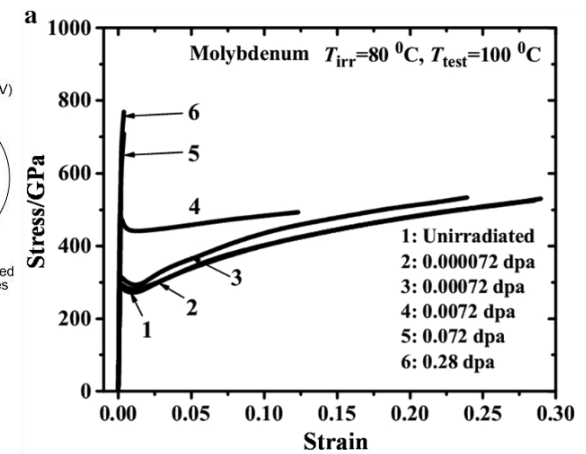


Fig. Irradiation hardening [3]

Importance of IASCC

- **Irradiation assisted stress corrosion cracking (IASCC)** describes a phenomenon where the interplay of environment, stress, and material, akin to conventional SCC, is hastened by radiation. This results in alterations in **corrosion resistance and mechanical properties**, **worsening SCC**.
- To assess the integrity of materials afflicted by IASCC, comprehending the **ramifications of neutron irradiation is paramount**.

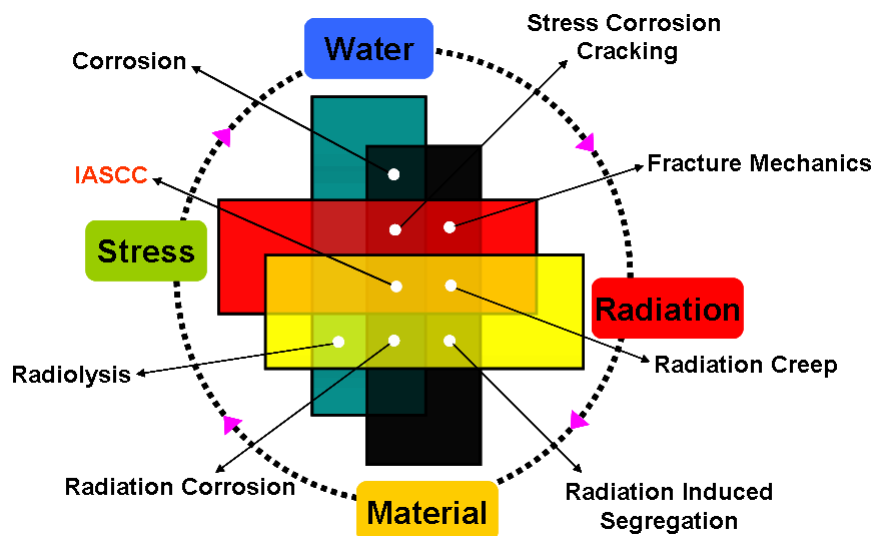


Fig. Irradiation Assisted Stress Corrosion Cracking [1]



Fig. IASCC crack observed in Baffle bolt [2]

The Cr Depletion caused by RIS

- RIS arises from variations in vacancy and interstitial concentrations induced by material diffusion rates.
- In stainless steel, chromium depletion hampers the development of the passive layer, rendering it prone to corrosion and serving as a critical factor to assess as a primary contributor to SCC.
- Chromium depletion initially occurs rapidly, according to a 2 dpa criterion, before decelerating and manifesting within a narrow range.

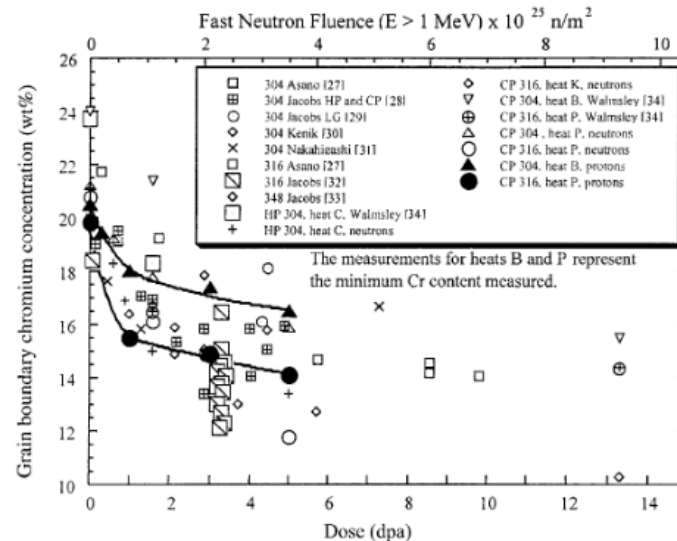


Fig. Cr composition as a function of dose for irradiated [1]

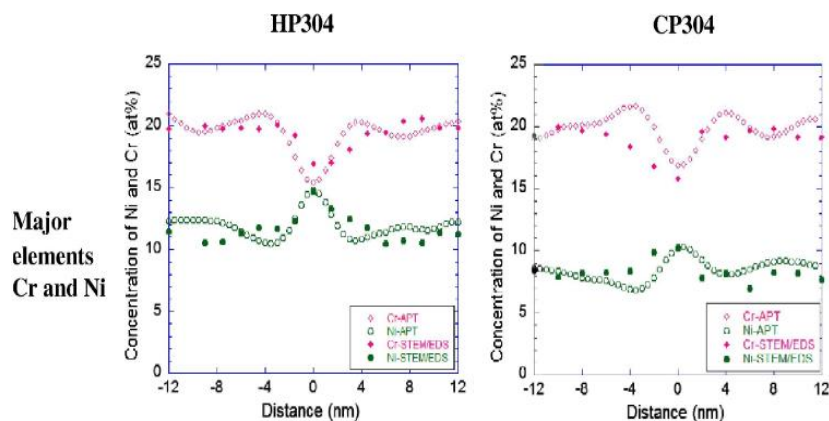


Fig. Cr composition as a function of dose for irradiated [2]

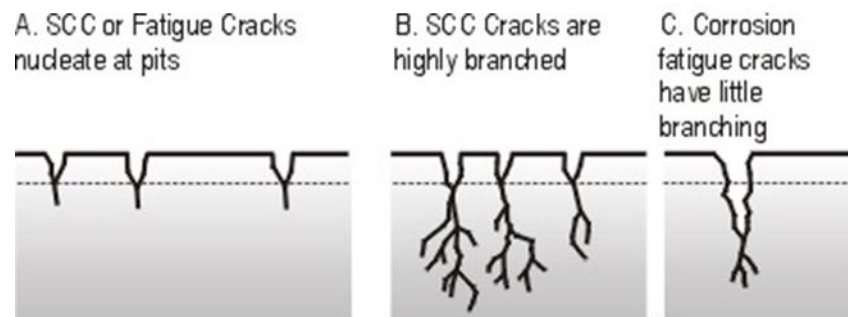


Fig. The mechanism of SCC caused by pitting formation

1. G.S. Was, J.T. Busby et al., *Journal of Nuclear Materials* (2002)
2. Z.Jiao, G.S. Was et al., *Acta Materialia* (2011)

Mechanical Properties by Irradiation Hardening

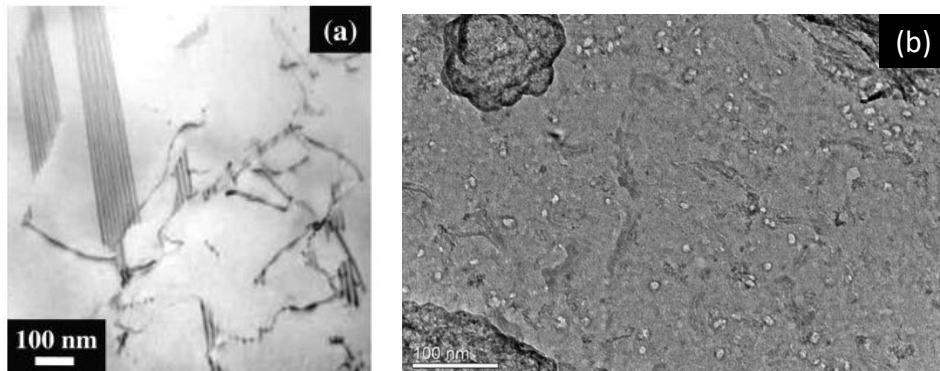


Fig. Investigation hardening due to microstructural changes
(a) dislocation loop, (b) He bubble [1], [2]

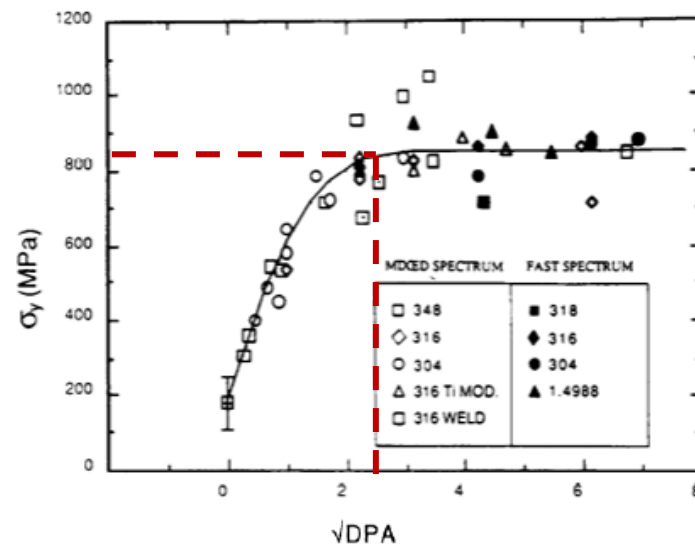
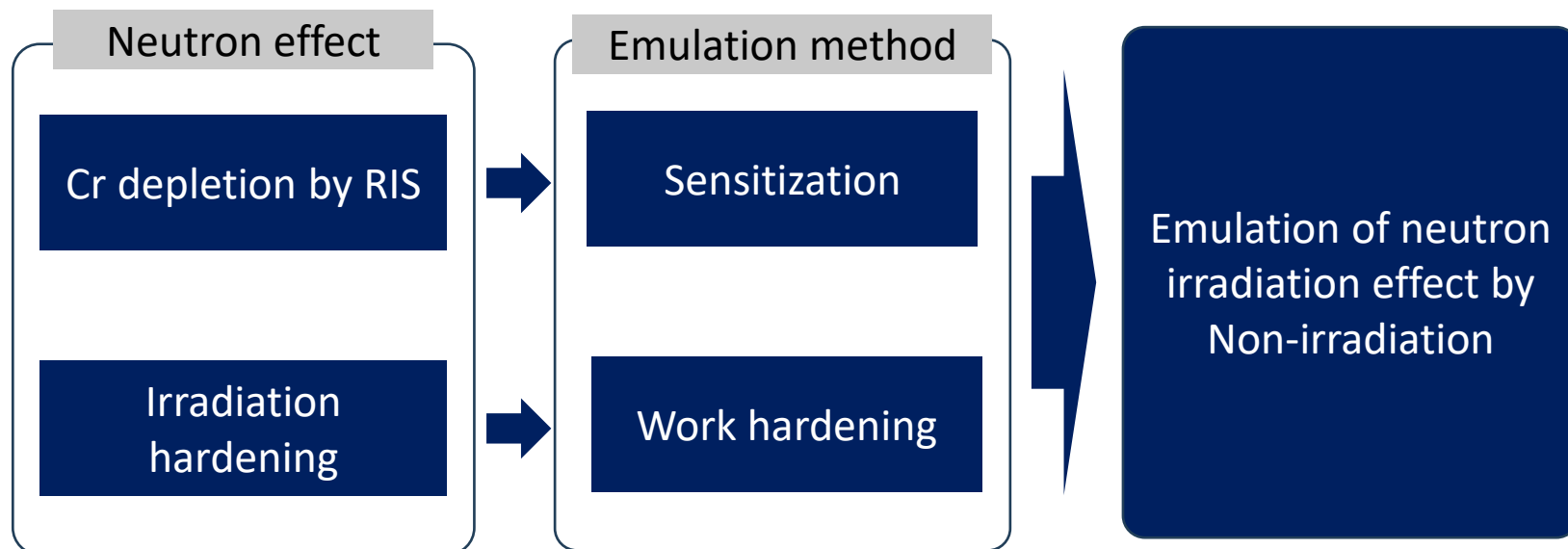


Fig. Mechanical property changes due to radiation [3]

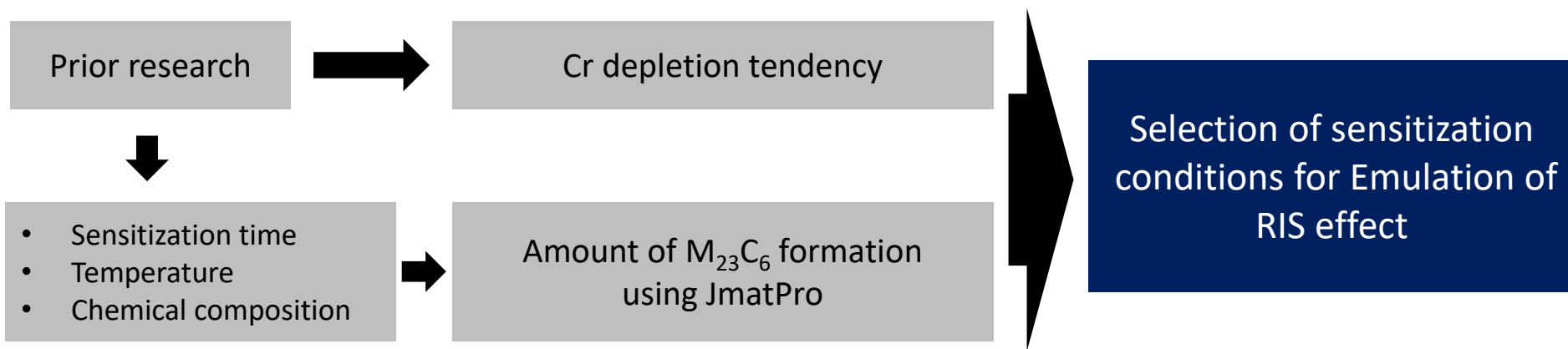
- Neutron irradiation induces hardening and embrittlement in materials through defect formation, resulting in elevated yield strength and reduced elongation.
- This hardening effect compromises fracture toughness, heightening the material's vulnerability to stress corrosion cracking.
- Upon surpassing 5 dpa of irradiation, the yield strength tends to stabilize, reaching around 800 MPa.

Objective

- Several methods for emulating neutron irradiation effects are under investigation, and I aim to propose a methodology for emulating neutron irradiation without actual irradiation.
- To emulate Cr depletion, sensitization heat treatment will be employed, while mechanical embrittlement will be induced through work hardening via rolling. The objective is to emulate mechanical strength reaching saturation at 5 dpa.

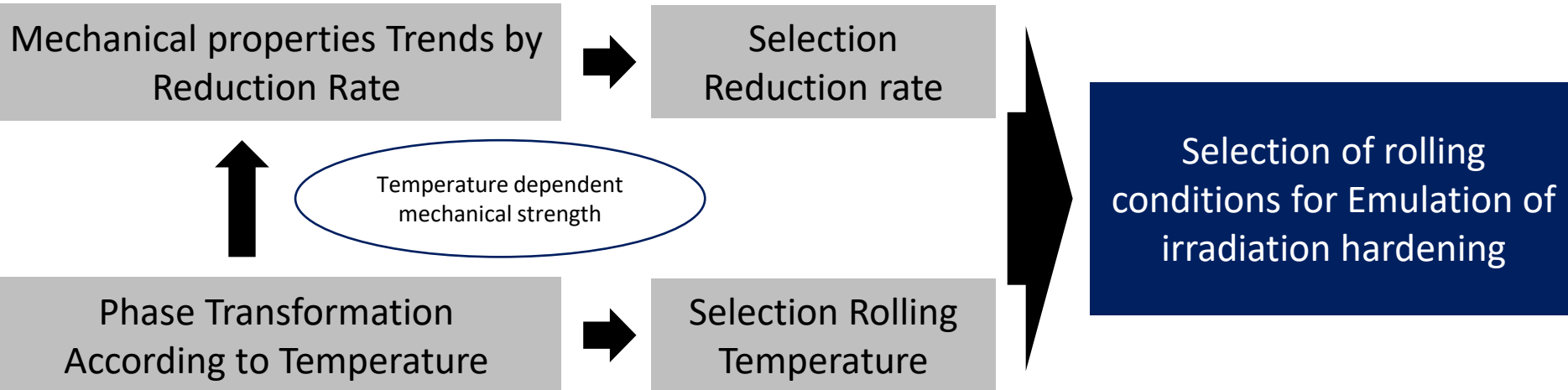


Prediction of Cr Depletion through Prior Research



- Analyze the trend of Cr depletion and establish sensitization temperatures by referencing phase diagrams.
- Validate the pattern of Cr depletion across different sensitization durations through existing literature to replicate Cr depletion accurately.
- Furthermore, utilize the thermodynamic computational simulation program JmatPro to confirm the formation of chromium carbides in SS 304 based on previous research.
- Compare the outcomes from both approaches to forecast the required heat treatment duration.

Prediction of Rolling Condition through Prior Research



- Analyzing hardenability trends based on rolling rate via literature review and adjusting hardenability and rolling rate accordingly.
- Confirming the amount of martensite phase formation at various temperatures during rolling.
- Investigating hardenability trends concerning temperature.
- Choosing the appropriate rolling temperature and rate.

Trend of Sensitization and Selection temperature

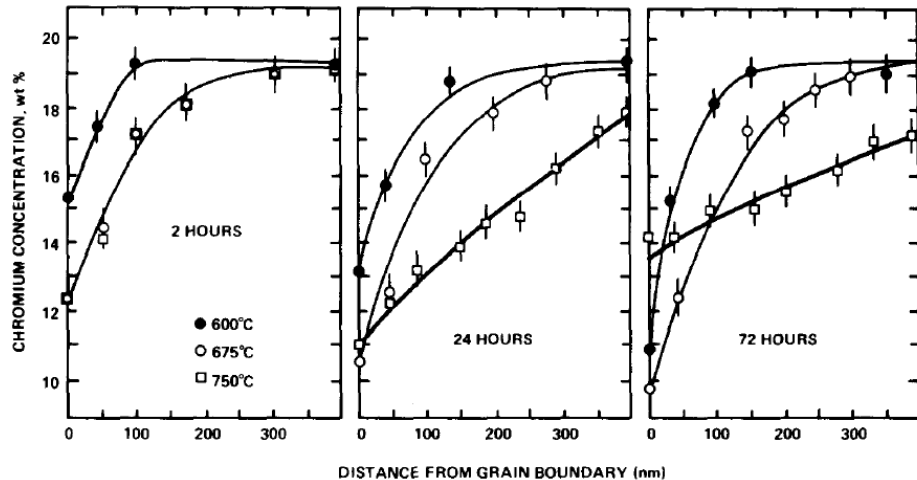


Fig. Heat treatment tendencies of SS304 [1]

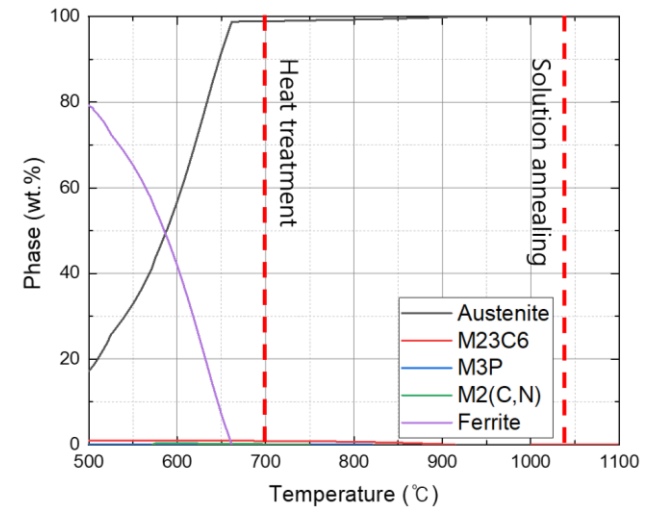


Fig. Phase diagram of SS304H [2]

- During heat treatment, Cr depletion initially experiences a substantial decrease, succeeded by an expansion in the depletion width and a reduction in the rate of depletion.
- Moreover, the extent of Cr depletion and the width of depletion are impacted by the heat treatment temperature, with a higher **sensitivity observed in the depletion width**.
- Given the **narrow Cr depletion width** observed in actual neutron irradiation, it's imperative to minimize the temperature.
- Austenitization is carried out at 1,040°C to **avert undesirable ferrite phase formation and diminish the Cr depletion width**, followed by sensitization at 700°C.

Selection of Sensitization Time

- Formation of $M_{23}C_6$ ranges from a **minimum of 1 hour to a maximum of 0.52 at.%**
- To achieve the formation of 0.52 at.% Cr carbide, approximately 12 hours of heat treatment are necessary. Accounting for other variables, this duration can be extended up to a **maximum of 24 hours**.

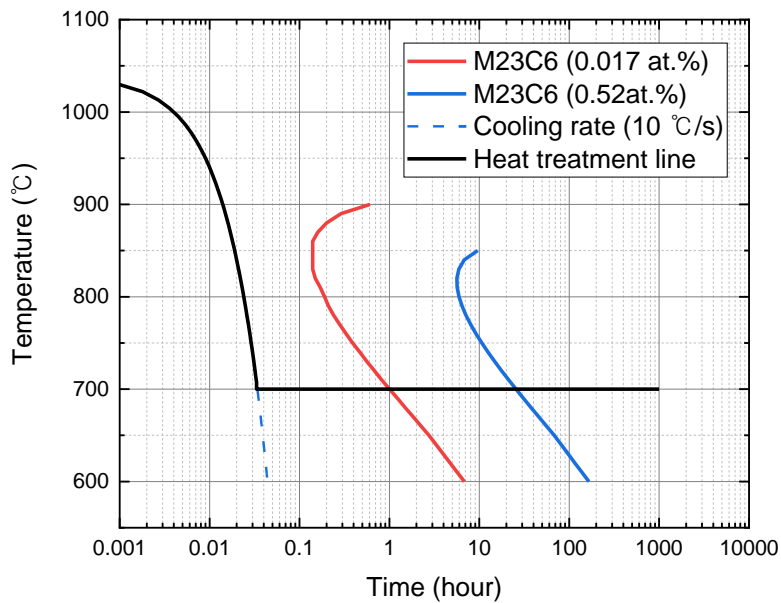


Fig. TTT diagram for $M_{23}C_6$ of SS 304H [5]

Table. In prior research papers, the trend of Cr changes in SS304 was compared with the calculation results of $M_{23}C_6$ using JmatPro [1], [2], [3], [4], [5].

Author	Temp. (°C)	Time (h)	$M_{23}C_6$ (at.%) [5]	ΔCr (wt.%)
Bruemmer [2]	600	9	0.0298	5.38
Bruemmer	600	57	0.2266	6.58
Bruemmer	600	100	0.3986	7.08
Bruemmer	625	25	0.17	7.28
Butler [1]	600	2	0.00645	3.8
Butler	600	2	0.00645	1.9
Butler	675	2	0.0401	6.5
Butler	675	2	0.0401	6
Butler	675	2	0.0401	5.5
Butler	675	2	0.0401	4.9
Bruemmer	700	1	0.0226	5.98
Bruemmer	700	10	0.3	7.08
Bruemmer	700	100	1.25	6.58
Ravindra [3]	700	2	0.0436	6.011
Ravindra	700	24	0.55	8.711
Ravindra	700	96	0.89	8.011
Trillo [4]	775	10	0.52	4.79
Bruemmer	800	10	0.85	3.88

Table. Test matrix in this paper

Annealing Temperature [°C]	1040
Annealing time	2 hour
Sensitization Temperature [°C]	700
Sensitization Time [hour]	1, 2, 3, 5, 7, 15, 24
Cooling	Water quenched

Carbide Distribution due to Heat Treatment

$A = Cr$ carbide area, $L =$ Grain boundary length

- Verifying the area of Cr carbides in relation to the length of grain boundaries over different heat treatment durations using SEM.
- It's important to note that Cr depletion cannot be confirmed via SEM. Quantitative evaluation of Cr depletion requires STEM.

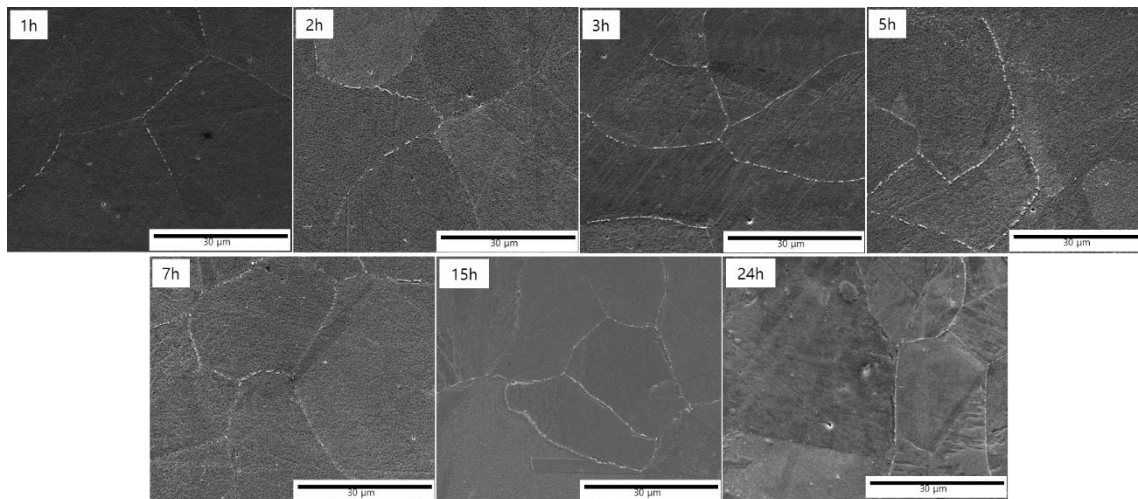


Fig. SEM images of SS304H by heat treatment durations

Table. In Prior research paper, trend of Cr depletion

Aging time (h)	1	2	3	5	7	15	24
Area/length(A/L)	0.0087	0.0296	0.0338	0.0380	0.0473	0.0506	0.0697

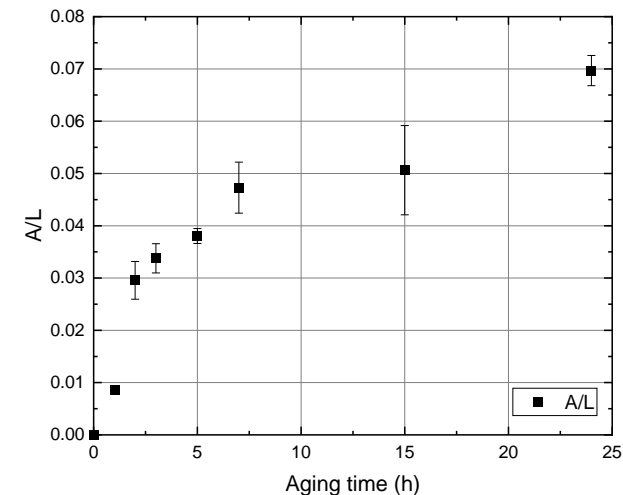


Fig. A/L value of SS304H by heat treatment durations

Cr Carbide Analysis using STEM

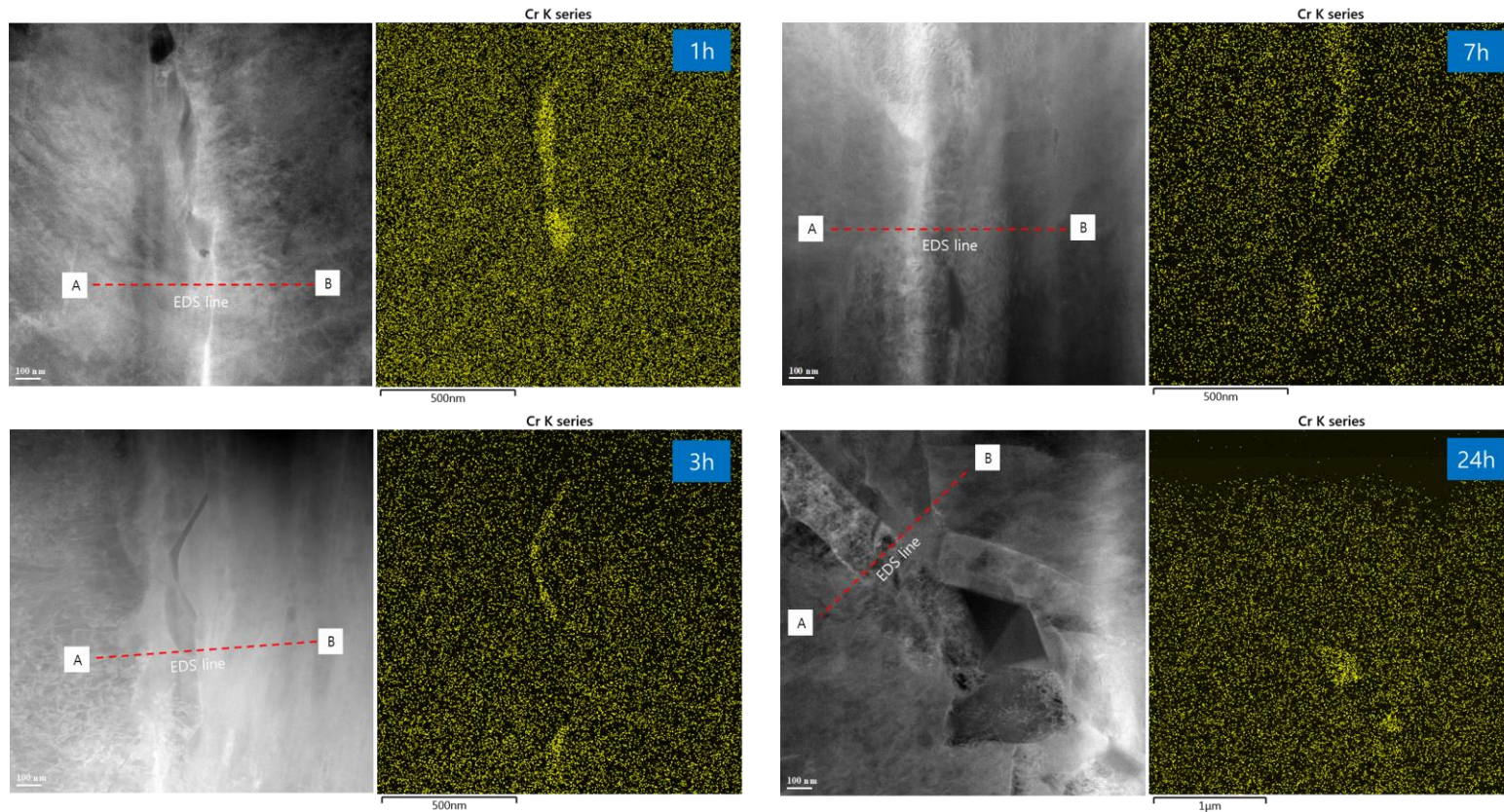


Fig. STEM image with EDS line and EDS map of Cr

- Conducting measurements of the EDS line along the grain boundary, maintaining a consistent distance (100nm to 150nm) from the Cr carbide.

Comparison of Cr Depletion Trends between Heat Treatment and RIS

- The variation of Cr experiences a sharp change initially with heat treatment time.
- After 7 hours, the variation of Cr stabilizes, while the **depletion width undergoes significant changes after 3 hours.**
- Comparison with actual 5 dpa neutron-irradiated materials indicates that the optimal **heat treatment duration is approximately 3 hours.**

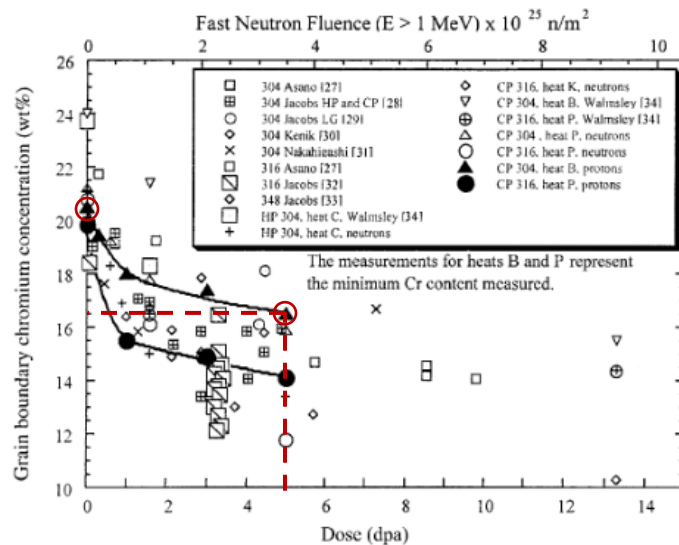


Fig. Cr depletion trend due to RIS [1]

Table. Trend of Cr concentration sensitization time

Heat treatment time (Hour)	Min. of Cr (wt.%)	Depletion width (nm)
1	14.76	150
3	12.2	180
7	11.42	250
24	11.14	300

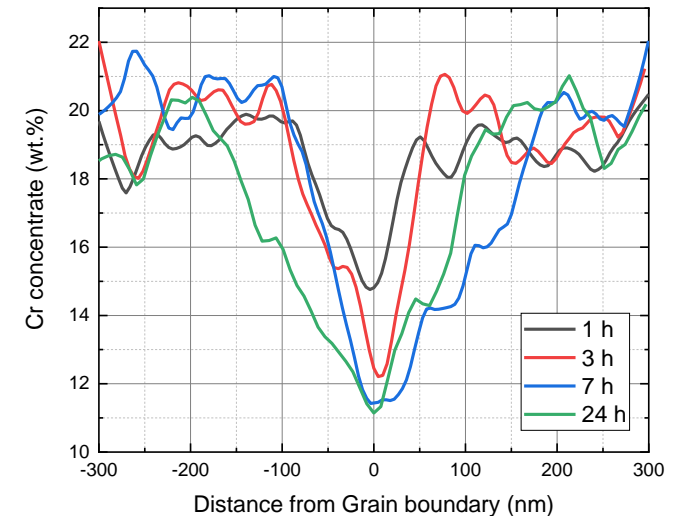


Fig. Trend of Cr concentration by heat treatment time

Methodology for Selecting Rolling Temperature

- Warm rolling is performed at high temperatures to mitigate undesired Martensite phase formation.
- However, during warm rolling, the increment in the original yield strength tends to be diminished, necessitating careful consideration.
- Our objective is to enhance the yield strength by conducting rolling at 200 °C. To account for temperature effects, we plan to execute approximately 10% more rolling compared to the existing reduction rate.

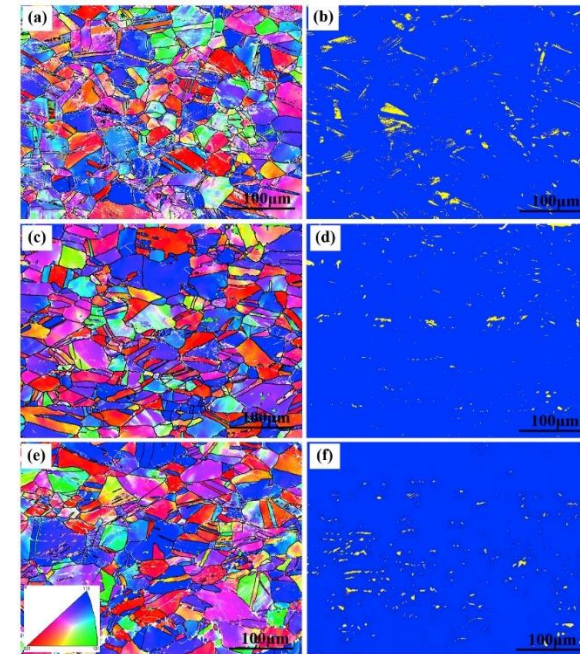


Table. Yield strength by hot rolling temperature (RD, reduction rate = 40%, SS316) [2]

Temperature (°C)	$\sigma_{0.2}$ (MPa)	UTS (MPa)	Elongation (%)
500	945	1080	10.5
700	875	960	11.5
900	720	820	20.0

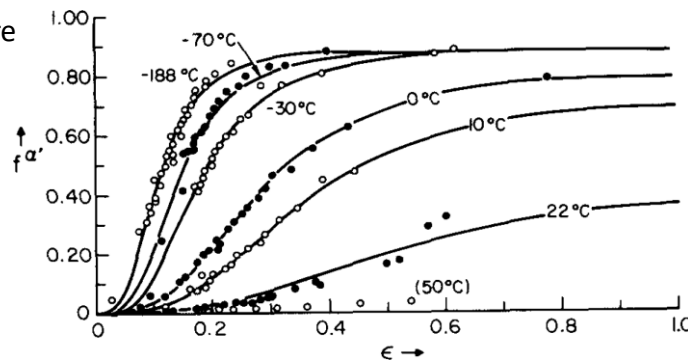


Fig. Martensite Fraction by Temperature [1]

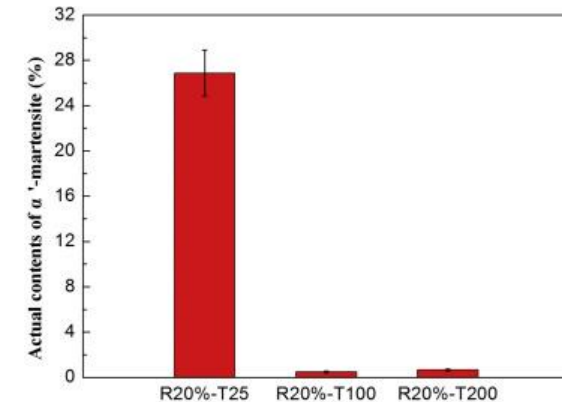


Fig. Martensite Fraction by temperature [3]

Methodology for Selecting Rolling Rate

- We emulate strain hardening by inducing defect formation through rolling.
- We attained the desired yield strength of **800 MPa with approximately 25% cold rolling.**
- Considering that the hardening effect diminishes as temperature rises

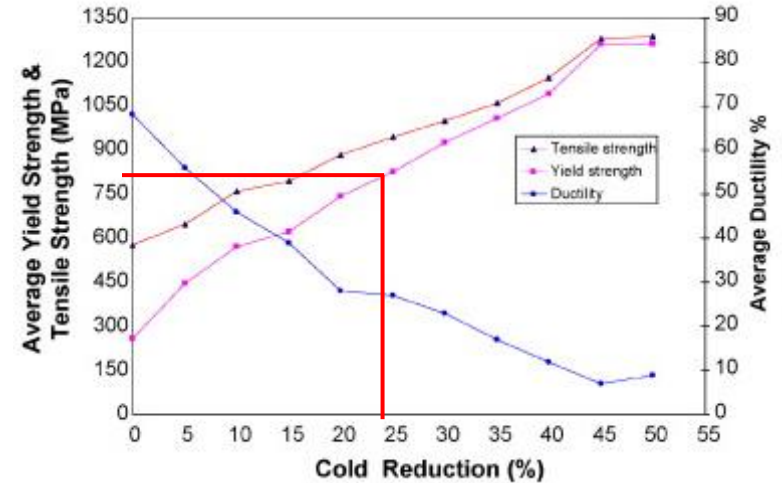


Fig. Yield strength by reduction rate (TD, Room Temperature, SS304) [1]

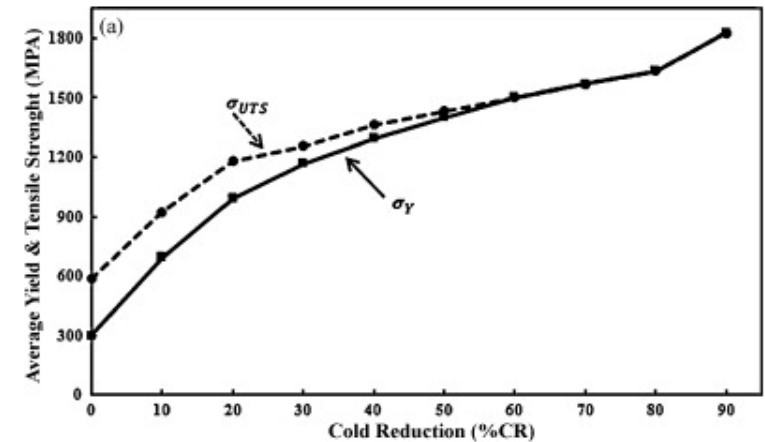


Fig. Yield strength by reduction rate (RD, 0°C, SS304) [3]

Table. Test matrix in this paper

Temperature [°C]	200
Reduction rate [%]	10, 20, 30

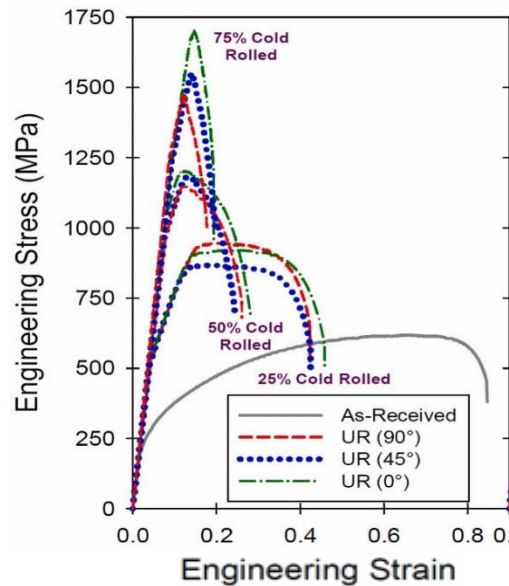


Fig. Stainless Steel 316 Tensile test graph [2]

The Tensile Test Results by Rolling

- As the reduction rate rises, the material's yield strength increases while elongation experiences a notable decrease, mirroring traits observed in neutron-irradiated materials.
- The irradiation hardening phenomenon at 5 dpa was discovered to closely resemble conditions at **approximately 800 MPa**, corresponding to around a **30% reduction rate**.

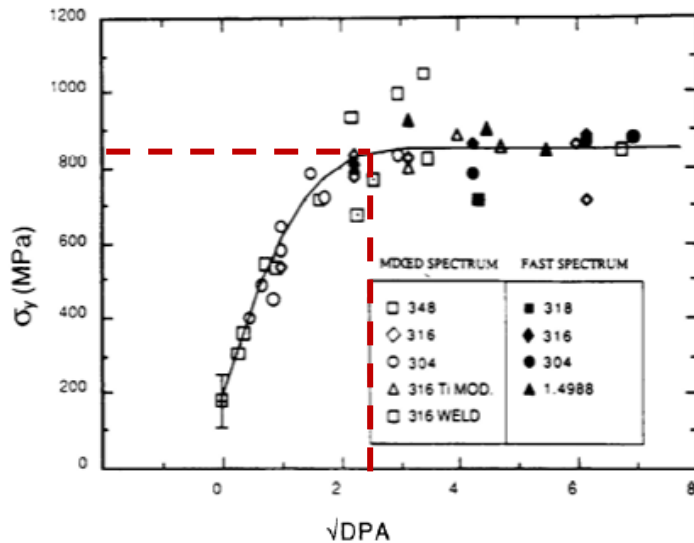


Fig. Mechanical property changes by irradiation [1]

Table. Mechanical property changes by rolling

Reduction rate	3hr, 0%	3hr, 10%	3hr, 20%	3hr, 30%
Yield strength (MPa)	233	443	568	789

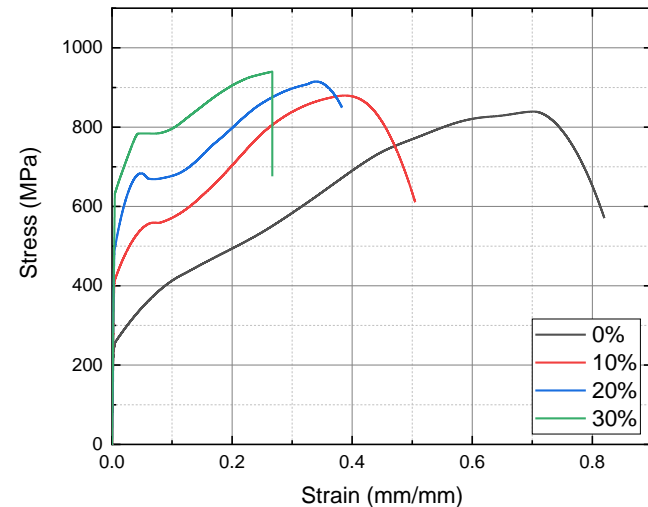


Fig. Mechanical property changes by rolling

Conclusion and Future work

Conclusion

- Neutron irradiation of stainless steel significantly contributes to SCC. **RIS and irradiation hardening play pivotal roles**, crucial in understanding IASCC.
- However, analyzing these factors entails risks and substantial costs. Therefore, our approach focuses on **emulating them using non-irradiated methods**, aiming to achieve a mechanical strength saturation equivalent to 5 dpa. This will be accomplished through **heat treatment to mimic RIS and work hardening via rolling**.
- Utilizing thermodynamic computational simulation programs, we determined optimal heat treatment conditions, concluding that conducting heat treatment at **approximately 700°C for about 3 hours** is most suitable. Similarly, **rolling at a 30% reduction rate at 200°C** was identified as the most appropriate method.

Future work

- Comparison of corrosion susceptibility according to microstructural changes between heat-treated and neutron-irradiated specimens
- Comparing Neutron-Irradiated Materials with SCC Susceptibility



Thank you
for your attention

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